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Flux pinning, granularity and the irreversibility line of the high- T_c superconductor $\text{HgBa}_2\text{CuO}_{4+x}$

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THE recent discoveries of superconductivity at 94 K in the single- CuO_2 -layer compound $\text{HgBa}_2\text{CuO}_{4+x}$ (Hg-1201)¹ and at 133 K in its two- and three-copper-layer analogues² (Hg-1212 and Hg-1223) have renewed interest in the search for new high-transition-temperature (high- T_c) superconductors. But whatever the T_c , high values of intergrain critical current density are required for large-scale applications; thus, the material must have good electromagnetic connectivity between grains (it must not be electromagnetically granular) and it should be able to maintain its current-carrying capacity in high magnetic fields (it should have a high irreversibility field, $H^*(T)$). Patil et al.³ surmised that the small CuO_2 layer spacing of Hg-1201 might yield a high $H^*(T)$. We report here that $H^*(T)$ is indeed significantly higher than for the two- and three-copper-layer Bi-Sr-Ca-Cu-O compounds (Bi-2212 and Bi-2223), but lower than for $\text{YBa}_2\text{Cu}_3\text{O}_7$ (Y-123). The low-temperature flux pinning is also strong. Like Y-123 , however, Hg-1201 is electromagnetically granular. A high degree of grain alignment will probably be necessary to remove this granularity; by analogy with the bismuth compounds, this is likely to be easier if the two-mercury-layer counterparts of Bi-22XY (Hg-2201 , -2212 or -2223) can be synthesized.

$\text{HgBa}_2\text{CuO}_{4+x}$ was prepared from stoichiometric mixtures of HgO and Ba_2CuO_3 . Ba_2CuO_3 was prepared by reacting BaO_2 (<95%) and CuO (<99.9%) at 900°C in oxygen for 24 hours using one intermediate grinding step. Storage, grinding, and mixing of Ba_2CuO_3 with HgO was done in an inert atmosphere. Pellets of the mixture were placed in an alumina crucible, which in turn was placed inside a quartz tube. This was evacuated, sealed, then placed in a stainless steel tube filled with sand to protect against possible explosion (which did not occur). The samples studied here were heated to 800 °C at 5 °C min⁻¹, then slowly cooled to room temperature over 10 hours. Samples heated at 845 °C exhibited sharper X-ray patterns for the 1201 phase but also showed free Hg, as also seen by Wagner et al.⁴.

For this reason we worked at 800 °C. Subsequent transmission electron microscopy (TEM) on the 800 °C samples showed that they too had some free Hg (or HgO) within them, the size of these particles being 10–30 nm. Our X-ray, scanning electron microscopy (SEM), TEM, and magnetization measurements showed that the proportion of 1201 phase was > 50%. After reaction, the samples were annealed at 200 °C for 2–6 hours in flowing oxygen. The grain size of all samples was ~ 5 µm and the grain shape was nearly equiaxed. Mercury-rich compounds were observed at a number of grain boundaries, though others appeared clean, when examined by back scatter SEM and TEM.

The essence of our results is shown in Figs 1–4. Figure 1a shows the a.c. susceptibility transition of the as-reacted and subsequently oxygenated samples in a zero-to-peak field of 0.1 mT. Both transitions are rather more abrupt than in the original report¹. The treatment in oxygen markedly raised the T_c onset from ~ 80 K to ~ 91 K. Both samples appeared electromagnetically well-connected under this small a.c. field and the sharp transition indicates the basic good quality of the Hg-1201 phase. Raising the a.c. field however, from 0.1 to 4 mT, produced a marked broadening of the transition (shown in Fig. 1b) consistent with electromagnetic granularity caused by the field breaking the intergrain shielding current loops. The d.c. magnetization was measured in a vibrating sample magnetometer (VSM) at fields up to 12 T from 4 to 80 K. The critical current density (J_{cm}) was calculated from the modified Bean expression⁵ for the magnetization hysteresis $\Delta M = J_{cm}a_2(1 - (a_2/3a_1))/10$, where $2a_1$, $2a_2$ are the cross sectional dimensions and $a_1 > a_2$. The whole sample cross-section (1.5 mm × 2 mm) was used to derive J_{cm} in the limit that all the current was flowing from grain to grain (intergrain). The grain size (~ 5 µm) was

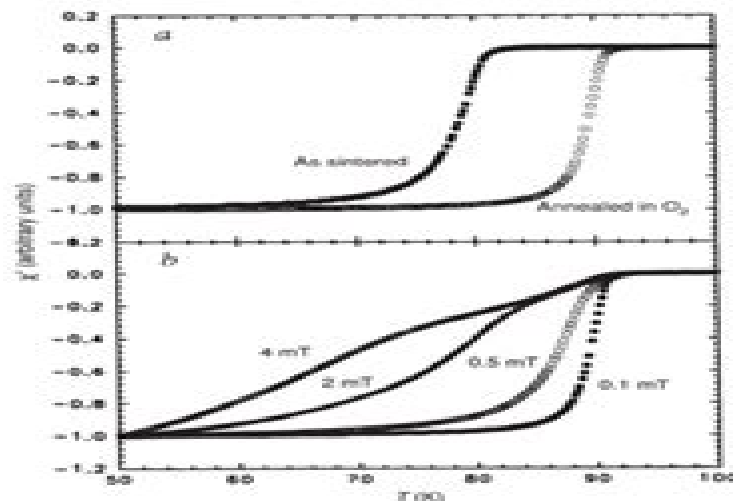


FIG. 1 a, Real component of the susceptibility (χ') for the as-sintered and oxygen-annealed samples at $H = 0.1$ mT. b, $\chi'(T)$ of the oxygen-annealed sample for a.c. fields between 0.1 and 4 mT.

Flux Pinning Phenomena In High Tc Superc

Daniël Cornelis van der Laan



Flux Pinning Phenomena In High Tc Superc:

Flux Pinning in Superconductors Teruo Matsushita, 2007-07-04 Superconductivity is now a considerable focus of attention as one of the technologies which can prevent environmental destruction by allowing energy to be used with high efficiency. The possibility of practical applications of superconductivity depends on the maximum current density which superconductors can carry, the value of losses which superconductors consume, the maximum magnetic field strength in which superconductors can be used etc. These factors are directly related to the flux pinning of quantized magnetic flux lines in superconductors. This book extensively describes related subjects from the fundamental physics of flux pinning to electromagnetic phenomena caused by flux pinning events which will be useful for anyone who wants to understand applied superconductivity. The Japanese edition was published for this purpose in 1994. Since then there has been significant progress in the research and development of high temperature superconductors. In particular, the new superconductor MgB₂ was discovered in 2001 followed by steady improvements in the superconducting properties necessary for applications. On the other hand, there are no essential differences in the flux pinning phenomena between these new superconductors and metallic superconductors. Hence the framework of the previous Japanese edition was kept unchanged while new description was added on these new superconductors in the English edition. In the following, the content of each chapter is briefly introduced.

Scientific and Technical Aerospace Reports, 1992 *Physics Briefs*, 1994 *International Aerospace Abstracts*, 1993 **Government Reports Annual Index**, 1991 *Superconducting Devices & Materials*, 1978 **Electrical & Electronics Abstracts**, 1988 *Almanach Österreichische Akademie der Wissenschaften*, 1998 Vols for 1854-19 include section Die feierliche Jahressitzung called Die feierliche Sitzung 1854-1914. Die statutenmäßige Jahressitzung 1915-21 published also separately 1848-99 *Metals Abstracts Index*, 1970 *Energy Research Abstracts*, 1979 **Government Reports Announcements & Index**, 1991 **Investigations on Flux Pinning and Creep in High-Tc Superconductors** Osvaldo F. Schilling Neto, 1991 *Vortex Physics and Flux Pinning* A. V. Narlikar, 2005 *Studies of High Temperature Superconductors* Volume 48 *Vortex Physics Flux Pinning and Connectivity in Polycrystalline High-temperature Superconductors* Daniël Cornelis van der Laan, 2004 *Special Issue on Flux Pinning and Other Related Phenomena in Type-II Superconductors*, 2007 **Flux Pinning in Two-dimensional High Temperature Superconductors**, 1996 Various types of high temperature superconducting single crystals TI 2223 Bi 2212 and Y 123 with oxygen contents between 6.4 and 6.64 were investigated by SQUID magnetometry. Both materials show very open_quotes low close_quotes lying irreversibility lines H_{llc} which follow an exponential law and rise rapidly below T_{T sub c} 0.5. High critical current densities are found only at low temperatures e.g. J_{sub c} is about 2.7 circ 10¹¹ Am⁻² at 1 T and 5 K in TI 2223. A comparison of the condensation energy provided by different types of defects with the thermal energy gives evidence for pinning of individual pancakes by the as-grown defect structure.

Theoretical Models of Flux Pinning and

Flux Motion in High- Tc Superconducting Oxides ,1991 Various issues involved in the development of phenomenological models of flux pinning and motion in high Tc oxides are discussed A simplified model is presented for the critical current density and is used to examine the question of whether flux flow results from an instability due to plasticity of the flux line array or from pin breaking Flux Pinning, Defect Analysis and Growth of High Temperature Superconducting Single Crystals Guang Yang,1993 Theoretical Models of Flux Pinning and Flux Motion in High Tc Superconducting Oxides ,1991
Spiral Growth, Flux Pinning and Peak Effect in Doped and Pure Bi-2212 High Temperature Superconducting Single Crystals Xiaolin Wang, University of Wollongong,1999

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